

Microstructural studies for rheology

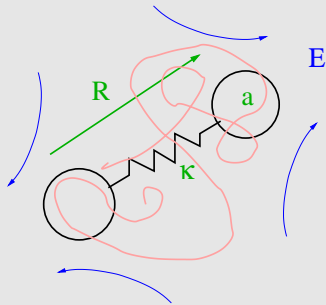
- ▶ Micro & macro views
- ▶ Einstein viscosity
- ▶ Rotations
- ▶ Deformations
- ▶ Interactions
- ▶ Polymers
- ▶ Others

Polymers

- ▶ Single polymer
 - ▶ Bead-and-spring model
 - ▶ Refinements
 - ▶ FENE-P constitutive equation
 - ▶ Unravelling a polymer chain
 - ▶ Kinks model
 - ▶ Brownian simulations
- ▶ Entangled polymers
 - ▶ rheology
 - ▶ Refinements
 - ▶ pom-pom

Bead-and-Spring model of isolated polymer chain

– simplest, only gross distortion, Kuhn & Kuhn 1945, Kramers 1946



- ▶ Flow distortion = Stokes drag = $6\pi\mu a(R \cdot \nabla U - \dot{R})$
 $a = \frac{1}{6}bN^{0.5} \rightarrow N^{0.6}$
- ▶ Resisted by entropic spring force = κR , $\kappa = \frac{3kT}{Nb^2}$

Hence

$$\dot{R} = R \cdot \nabla U - \frac{1}{2\tau} R \quad \text{with} \quad \tau = 0.8kT / \mu(N^{1/2}b)^3$$

Bead-and-Spring model of isolated polymer chain 2

- ▶ Adding Brownian motion of the beads: $A = \langle RR \rangle$

$$\overset{\nabla}{A} \equiv \frac{DA}{Dt} - A \cdot \nabla U - \nabla U^T \cdot A = -\frac{1}{\tau} \left(A - \frac{Nb^2}{3} I \right)$$

$$\sigma = -pI + 2\mu E + n\kappa A$$

with n number of chains per unit volume.

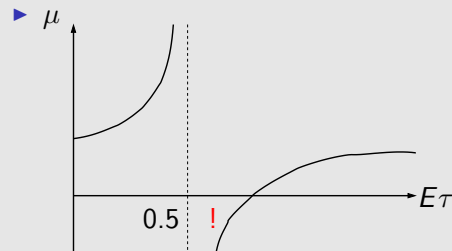
- Oldroyd-B constitutive equation with UCD time derivative $\overset{\nabla}{A}$

Rheological properties

Shear

- ▶ $\mu = \text{constant}$, $N_1 \propto \gamma^2$, $N_2 = 0$.
- ▶ Distortion xy : $a\gamma\tau \times a$

Extension



- ▶ Distortion $\propto e^{(2E - \frac{1}{\tau})t}$
- ▶ For TDR: small shear and large extensional viscosities

Refinements

1. (boring) **Spectrum** of internal modes: Rouse 53, Zimm 56 with pre-averaged hydrodynamics
2. (boring) **Polydisperse** molecular weights
3. (important) **Finite extensibility** – to stop infinite growth $\propto e^{(2E - \frac{1}{\tau})t}$
 - ▶ Nonlinear spring force – inverse Langevin law

$$F(R) = \frac{kT}{b} \mathcal{L}^{-1} \left(\frac{R}{Nb} \right) \quad \text{with} \quad \mathcal{L}(x) = \coth x - \frac{1}{x}$$

- ▶ F.E.N.E approximation

$$F(R) = \frac{kT}{Nb^2} \frac{R}{1 - R^2/L^2} \quad \text{with fully extended length} \quad L = Nb$$

- ▶ FENE-P closure

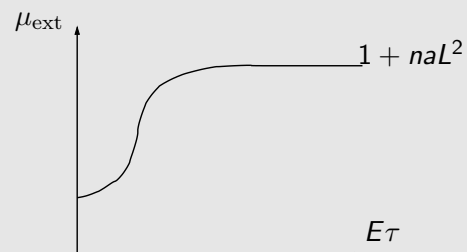
$$\langle RR / (1 - R^2/L^2) \rangle = \langle RR \rangle / (1 - \langle R^2 \rangle / L^2)$$

but “molecular individualism”

FENE-P constitutive equation

$$\overset{\nabla}{A} = -\frac{1}{\tau} \frac{L^2}{L^2 - \text{trace} A} \left(A - \frac{a^2}{3} I \right)$$

$$\sigma = -pI + 2\mu E + n\kappa \frac{L^2}{L^2 - \text{trace} A} A$$



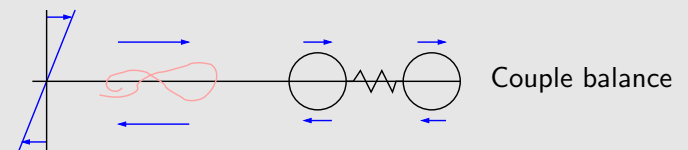
More refinements

4. Nonlinear bead friction

Hydrodynamic drag increase with size $6\pi\mu(a \rightarrow R)$

$$\mu_{\text{ext}} = 1 + nL^3 \quad \text{and hysteresis}$$

5. Rotation of the beads – simple shear not so simple



$$\text{Affine } \overset{\nabla}{A} \rightarrow \text{non-affine } \overset{\circ}{A} - \frac{\text{trace} A}{3 + \text{trace} A} (A \cdot E + E \cdot A)$$

inefficiency of straining

One more refinement

6. **Dissipative stress** – nonlinear internal modes
 Simulations show growing stretched segments

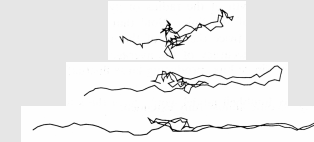
$$\text{segment length} \propto \frac{R^2}{L}, \quad \text{number} \propto \frac{L^2}{R^2}, \quad \text{dissipation} \propto \frac{R^4}{L}$$

$$\sigma = -pI + 2\mu \left(1 + n \frac{(\text{trace } A)^2}{L} \right) E + n\kappa \frac{L^2}{L^2 - \text{trace } A} A$$

Good for contraction flows

Unravelling a polymer chain in an extensional flow

Simulation of chain with $N = 100$ in uni-axial straining motion at strains $Et = 0.8, 1.6, 2.4$.

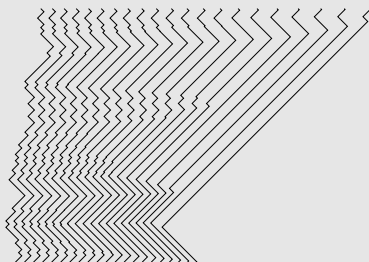


- ▶ Growing stretched segments
- ▶ Two ends not on opposite sides

Simplified 1D 'kinks' model

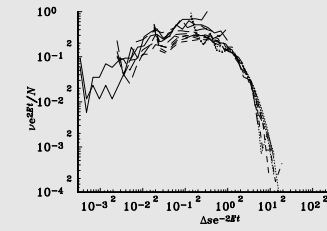
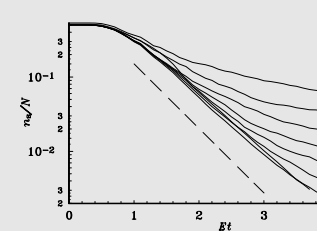
- ▶ $t = 0$: 1D random walk, N steps of ± 1
- ▶ $t > 0$: floppy inextensible string in $u = Ex$
- ▶ arc lengths satisfy

$$\dot{s}_i = \frac{1}{4} E (-s_{i+1} + 2s_i - s_{i-1})$$



- ▶ Large gobble small

Kinks model 2



Number of segments $n(t)$

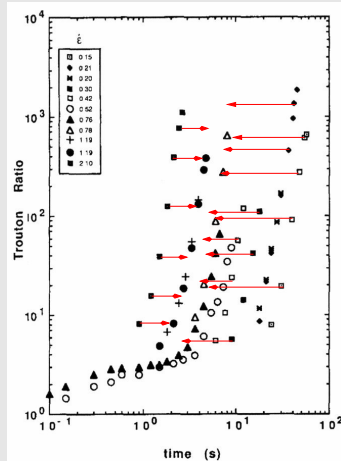
Distribution of lengths $l(t)$
 scaled by e^{2Et}

Scalings

$$\begin{cases} nl = N \\ \sqrt{nl} = R = \sqrt{N} e^{Et} \end{cases} \longrightarrow \begin{cases} n = N e^{-2Et} \\ l = e^{2Et} \end{cases}$$

Evidence of a dissipative stress

Original data of Sridhar, Tirtaatmadja, Nguyen & Gupta 1991 plotted as viscosity as function of time



Replotted a function of strain = strain-rate × time

Improved algorithms for Brownian simulations

1. **Mid-point time-stepping** avoids evaluating $\nabla \cdot \mathbf{D}$
Keep random force fixed in time-step, but vary friction
2. **Replace very stiff (fast) bonds** with rigid + correction potential

$$-kT \nabla \ln \sqrt{\det M^{-1}} \quad \text{with} \quad M^{-1ab} = \sum_i m_i^{-1} \frac{\partial g^a}{\partial \mathbf{x}_i} \cdot \frac{\partial g^b}{\partial \mathbf{x}_i}$$

where rigid constraints are $g^a(\mathbf{x}_1, \dots, \mathbf{x}_N) = 0$ and stiff spring energy $\frac{1}{2} |\nabla g^a|^2$

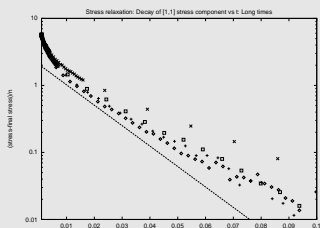
3. **Stress by subtraction** of large $\Delta t^{-1/2}$ term with zero average

$$\frac{1}{2}(x^n + x^{n+1})f^n \longrightarrow \frac{1}{2}\Delta x^n f^n$$

Grassia, Nitsche & H 95

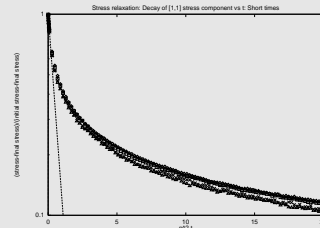
Relaxation of fully stretched chain

Long times - Rouse relaxation

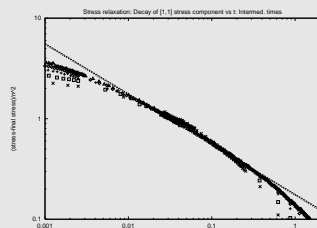


σ/N vs t/N^2 (Rouse)

Short times finite



$\sigma/\frac{1}{3}N^3$ vs N^2t



Intermediate times

$\sigma \sim kTN^2t^{-1/2}$

Constitutive equation – options

$$\begin{aligned} \nabla \cdot \mathbf{A} &= -\frac{1}{h\tau} f(\mathbf{A} - \mathbf{I}) \\ \boldsymbol{\sigma} &= -p\mathbf{I} + 2\mu\mathbf{E} + Gf\mathbf{A} \end{aligned}$$

- ▶ Oldroyd B $f = 1$
- ▶ FENE-P $f = L^2/(L^2 - \text{trace } \mathbf{A})$
- ▶ Nonlinear bead friction $h = \sqrt{\text{trace } \mathbf{A}/3}$
- ▶ New form of stress

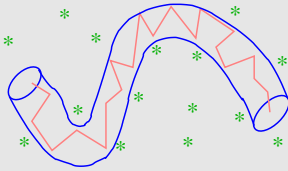
$$\boldsymbol{\sigma} = -p\mathbf{I} + 2\mu\mathbf{E} + 2\mu_d(\mathbf{A} : \mathbf{E})\mathbf{A} + G\sqrt{\text{trace } \mathbf{A}}\mathbf{A}$$

- ▶ Last term for finite stress when fully stretched
- ▶ μ_d term ($\propto N^{-1/2}$) for enhanced dissipation

Good for positive pressure drops and large upstream vortices in contraction flows.

Reptation model of De Gennes 1971 – often reformulated

Chain moves in tube defined by topological constraints from other chains.



Chain disengages from tube by diffusing along its length

$$\tau_D = \frac{L^2}{D = kT/6\pi\mu L} \propto M^3$$

Modulus $G = nkT \rightarrow \mu^* = G\tau_D \propto M^3$ (expts $M^{3.4}$)

Diffusion out of tube

At later time:



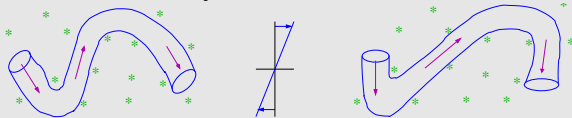
Fraction of original tube surviving

$$\sum_n \frac{1}{n^2} e^{-n^2 t / \tau_D}$$

Diffusion gives linear viscoelasticity $G' \propto \omega^{1/2}$

Doi-Edwards rheology 1978

Deformation of the tube by a shear flow.



Unit segments of the tube \mathbf{u} aligned by flow:

$$\mathbf{u} \rightarrow \mathbf{A}\mathbf{u} \text{ with Finger tensor } \mathbf{A}$$

Stress

$$\sigma(t) = n \int_0^\infty \sum_p \frac{1}{p^2} e^{-p^2 s / \tau_D} N_{\text{segments}} \frac{3kT}{a} a \left\langle \frac{\mathbf{A}^* \mathbf{u} \mathbf{A}^* \mathbf{u}}{|\mathbf{A}^* \mathbf{u}|^2} \right\rangle ds$$

surviving tube segment tension

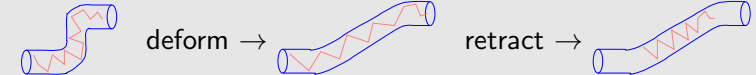
with relative deformation $\mathbf{A}^* = \mathbf{A}(t)\mathbf{A}^{-1}(t-s)$.

A BKZ integral constitutive equation

Problem maximum in shear stress

Refinements

1. Chain retraction



Chain returns in Rouse time to natural length \rightarrow loss of segments

2. Chain fluctuations

3. Other chains reptate \rightarrow release topological constraints

“Double reptation” of Des Cloiseaux 1990. bimodal blends

4. 2 & 3 give $\mu \propto M^{3.4}$

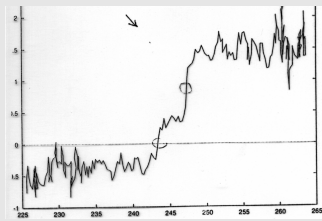
5. Advected constraint release Marrucci 1996

$$\frac{1}{\tau_D} \rightarrow \frac{1}{\tau_D} + \beta \nabla \cdot \langle \mathbf{u}\mathbf{u} \rangle$$

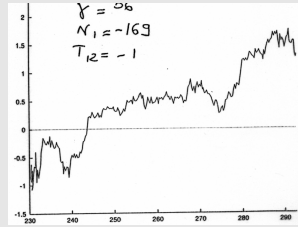
6. Flow changes tube volume or cross-section

Chain trapped in a fast shearing lattice

Lattice for other chains



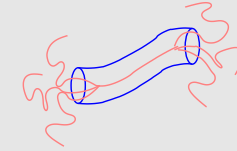
more shear →



central section pulling chain out of arms → high dissipative stresses

Ianniruberto, Marrucci & H 98

Branched polymers – typical in industry



Very difficult to pull branches into central tube

$$\mu \propto \exp(M_{\text{arm}}/M_{\text{entangle}})$$

Pom-Pom model of Tom McLeish and Ron Larson 1999

$$\text{Stress: } \sigma = G\lambda^2 \mathbf{S}$$

$$\text{Orientation: } \mathbf{S} = \mathbf{B}/\text{trace}\mathbf{B} \quad \overset{\nabla}{\mathbf{B}} = -\frac{1}{\tau_O}(\mathbf{B} - \mathbf{I})$$

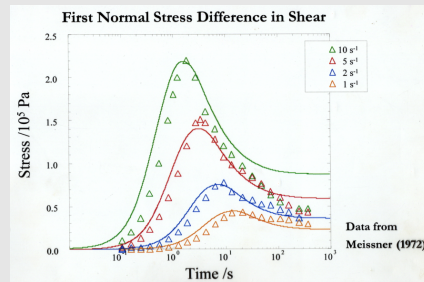
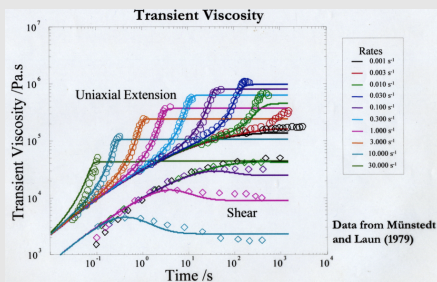
$$\text{Stretch: } \dot{\lambda} = \nabla u : \mathbf{S} - \frac{1}{\tau_S}(\lambda - 1) \quad \text{while } \lambda < \lambda_{\text{max}}$$

with $\tau_O = \tau_{\text{arm}}(M_C/M_E)^3$ and $\tau_S = \tau_{\text{arm}}(M_C/M_E)^2$ and $\tau_{\text{arm}} \cong \exp(M_{\text{arm}}/M_E)$ where $M_C = M_{\text{crossbar}}$ and $M_E = M_{\text{entanglement}}$.

Test of Pom-Pom model – Blackwell 2002

Fit: Linear Viscoelastic data and Steady Uni-axial Extension.

Predict: Transient Shear and Transient Normal Stress



IUPAC-A data Muntedt & Laun (1979)

Other microstructural studies

- ▶ Electro- and Magneto- rheological fluids
- ▶ Associating polymers
- ▶ Surfactants - micells
- ▶ Aging materials
- ▶ GENERIC
- ▶ Modelling 'Molecular individualism' and closure problems